



# Damage Models for SiC/SiC CMCs

**Peter Foster**, Adam Thompson, Riccardo Manno, Bassam El Said, **Luiz Kawashita**, Giuliano Allegri, Stephen Hallett

Supported by



# Background & Motivation

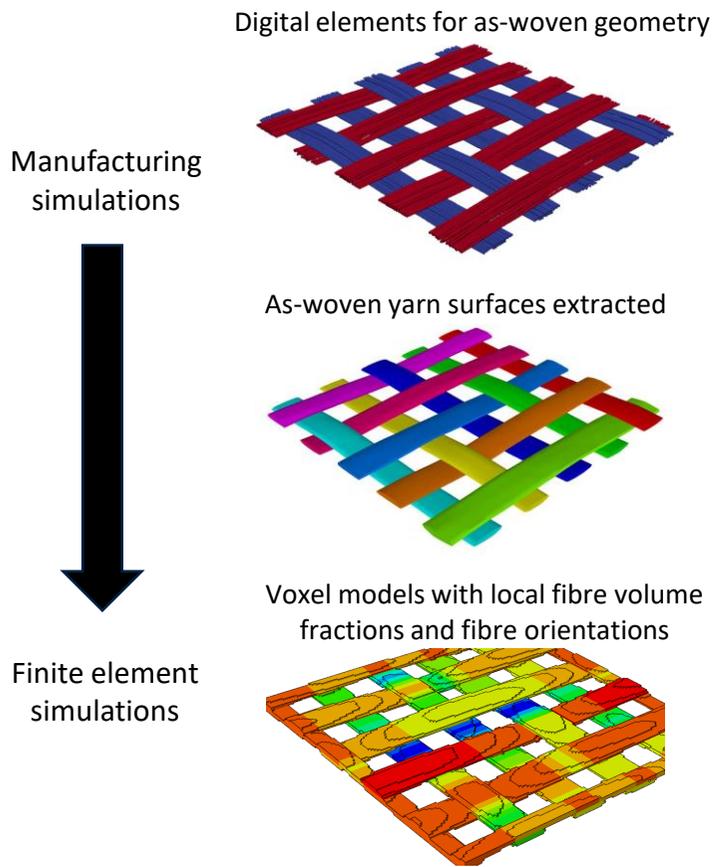
Questions being addressed:

1. Can the BCI models developed for woven **OMCs** be adapted for **SiC/SiC CMCs**?
2. How can the **complex damage mechanisms** of these materials be modelled efficiently and effectively?
3. At which **length scales** should these damage mechanisms be modelled?
4. Can **suitable assumptions** be made to compensate for the difficulties in characterising the various constituents and damage mechanisms?

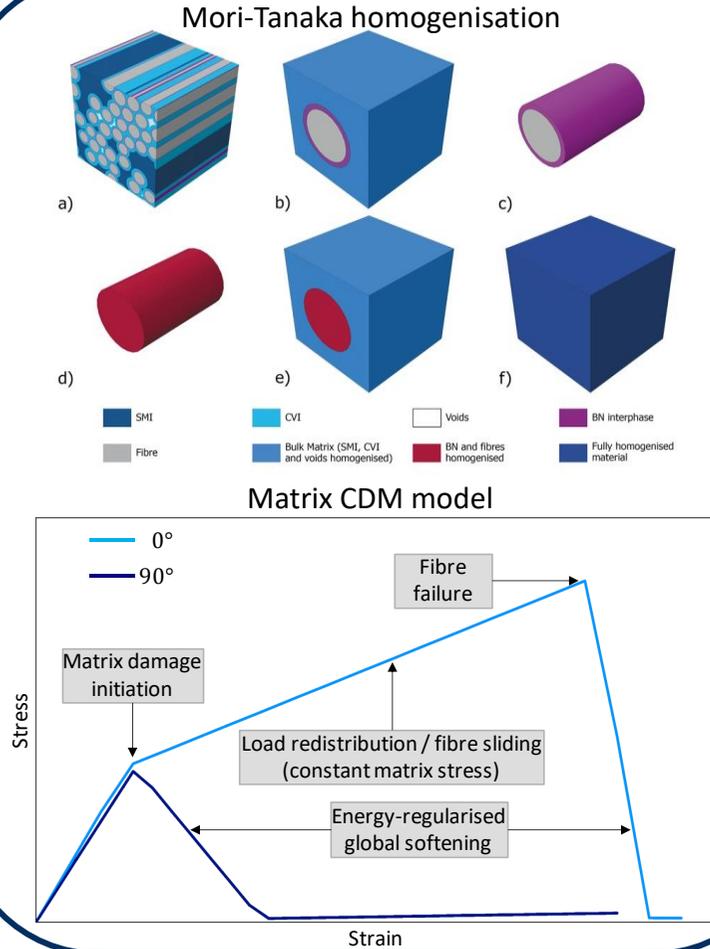


# BCI / UTC Meso-scale Modelling Framework

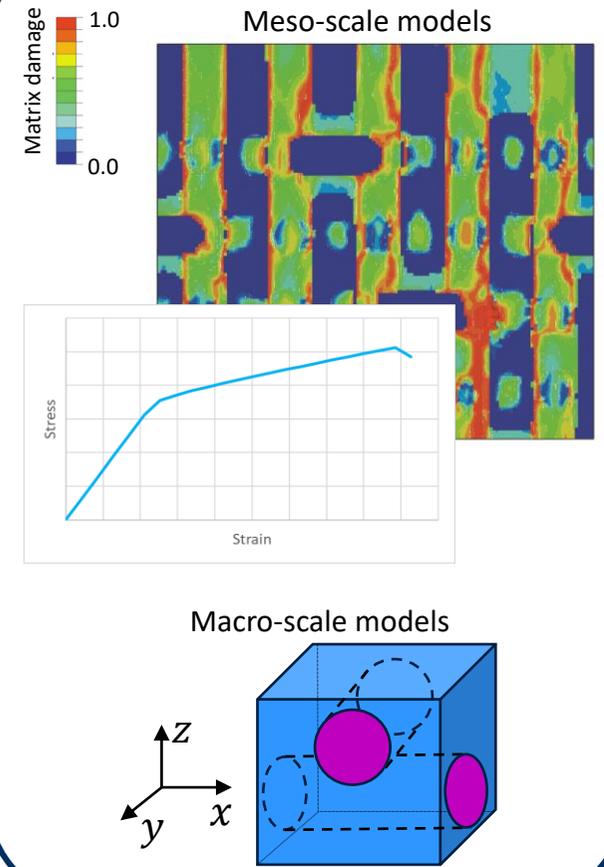
## Material Architecture



## Material Models

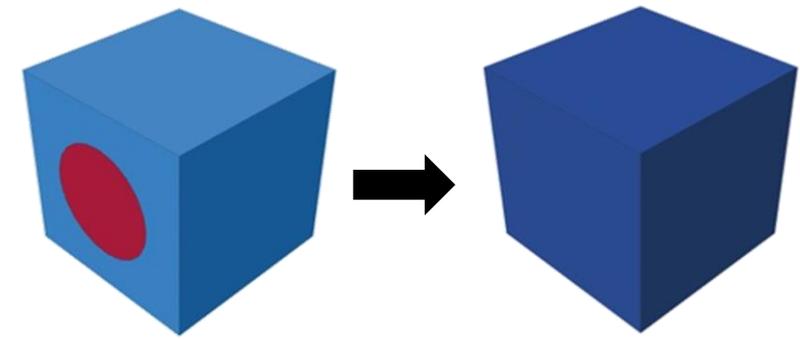


## Results



# Mori-Tanaka Homogenisation

- Homogenise an inclusion (**i**) with its surrounding medium (**m**) to get a global stiffness tensor  $C_{\text{homog}}$
- Strain concentration tensor  $A_i$  defines the relationship between globally applied strains and local constituent strains
- Required constituent data:
  - Elastic properties
  - Volume fraction
  - Shape of inclusion



*Inclusion and matrix*

*Homogenised model*

Strain concentration tensor of inclusion:

$$A_i = [I + \boxed{E} C_m^{-1} (C_i - C_m)]^{-1}$$

↑  
Eshelby tensor  
(inclusion 'shape')

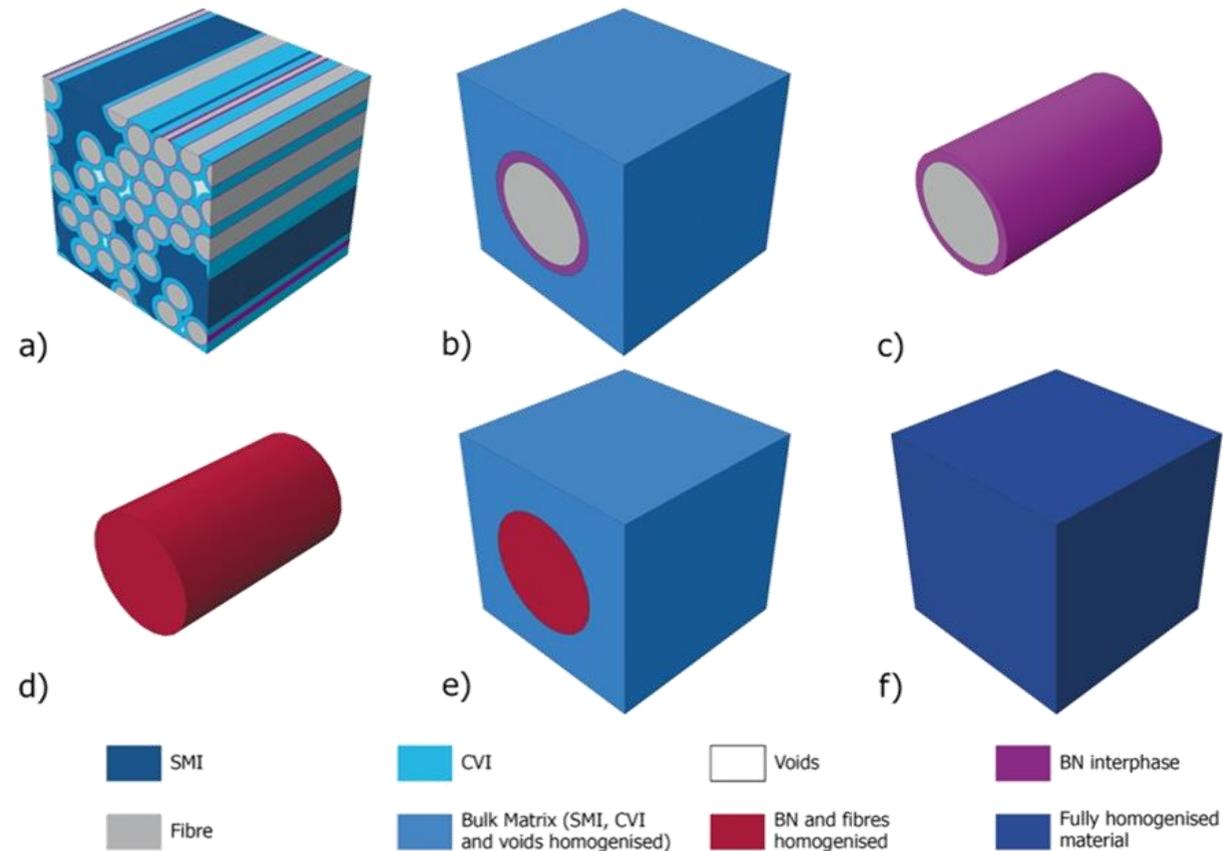
Homogenised stiffness tensor:

$$C_{\text{homog}} = C_m + \boxed{V_i} [A_i (C_i - C_m)]$$

↑  
Volume fraction  
of inclusion

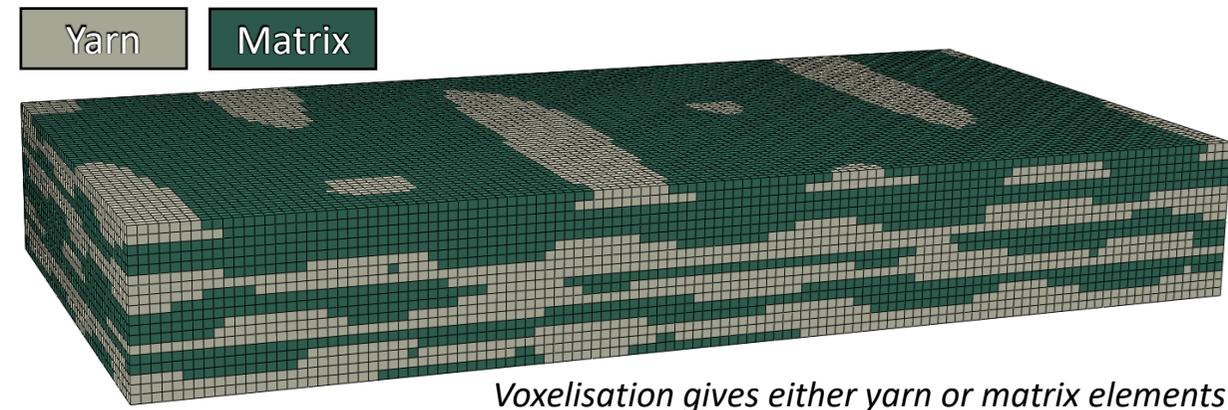
# Meso-scale Models of Woven SiC/SiC CMCs

## Mori-Tanaka homogenisation of CMC yarn



## Meso-scale material architecture

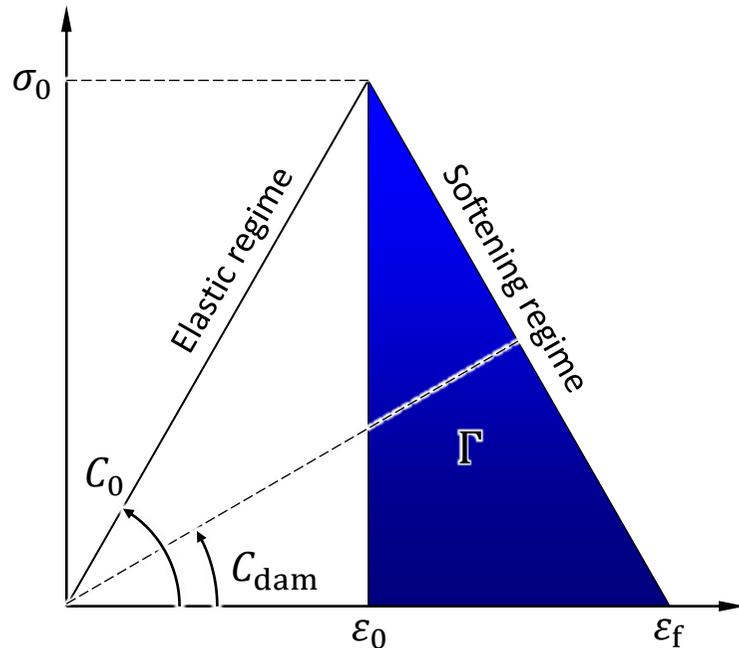
- BCI weaving and compaction models (SimTex) adapted for SiC/SiC CMCs
- Result: meso-scale voxel models containing **homogenised yarn** and **matrix** phases including local fibre volume fraction and fibre orientation



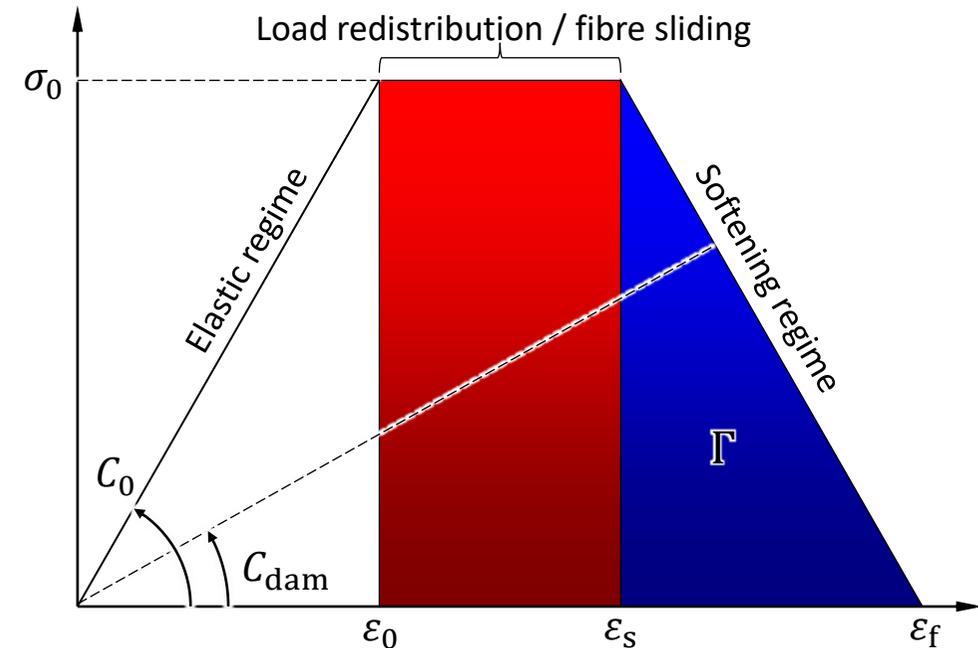
# Continuum Damage Mechanics (CDM) for SiC Matrix

- Energy-regularised CDM implemented for SiC matrix in (1) 'pure matrix' regions and (2) matrix constituent within yarns
- Load redistribution / fibre sliding after matrix damage modelled by a **constant matrix stress** assumption (until the fibre failure criterion is met)

Pure matrix loading / Yarn loaded transversally

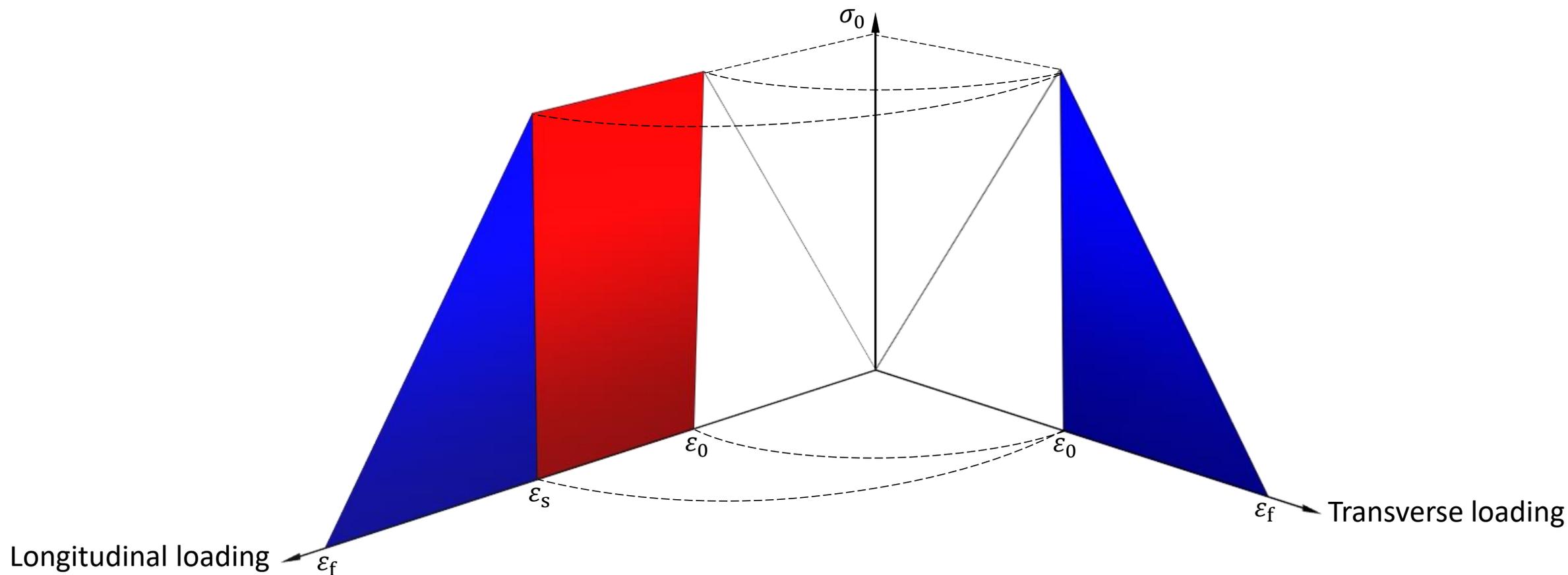


Yarn loaded longitudinally



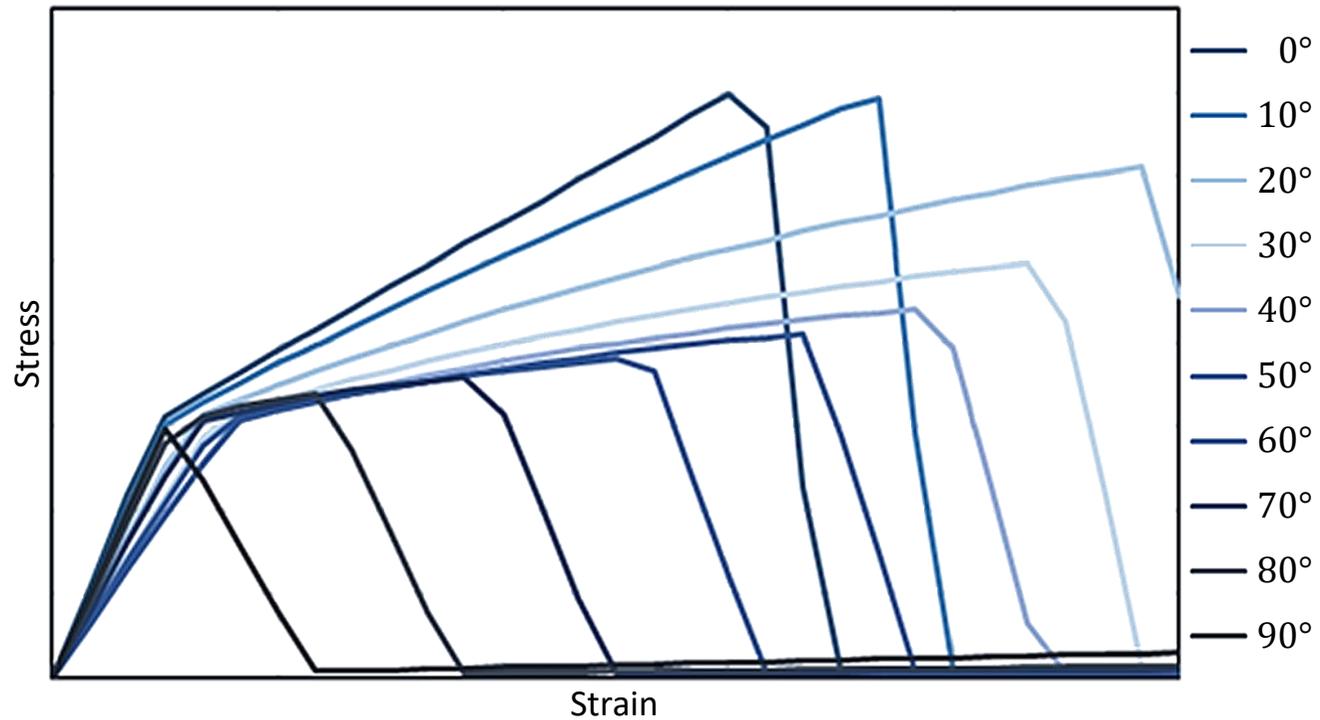
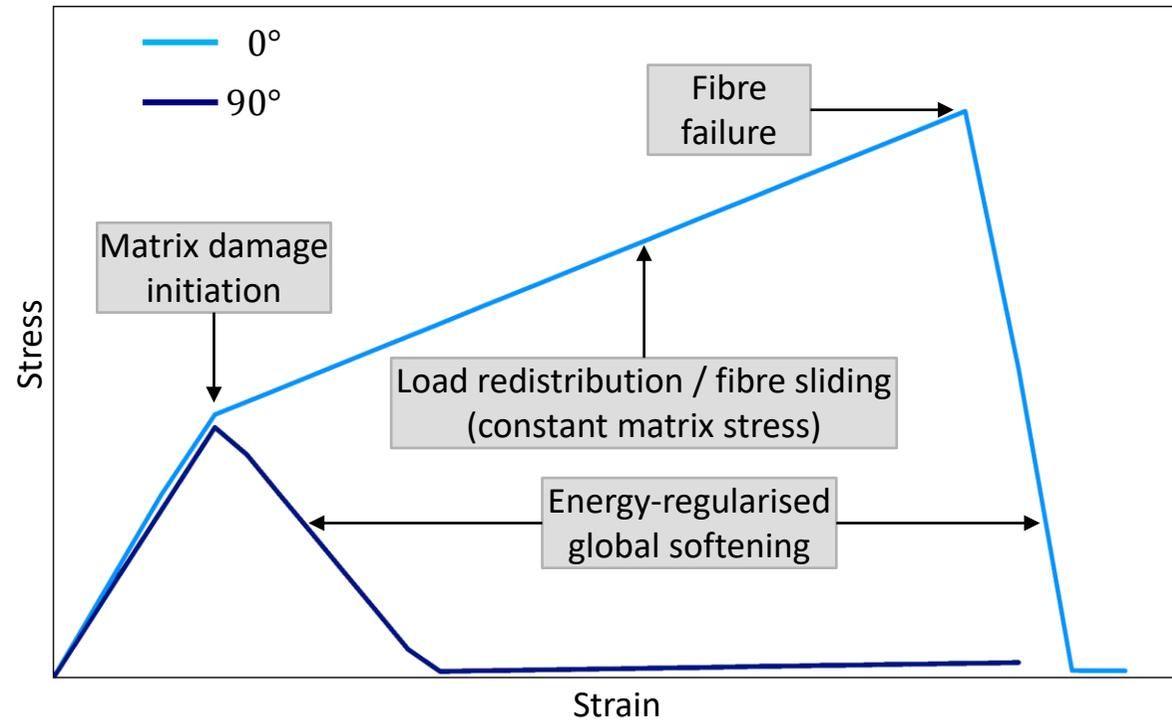
# CDM for Matrix Phase Within Yarns

- The damage behaviour of yarns is interpolated between the longitudinal and transverse loading cases based on the local fibre orientation and strain field:

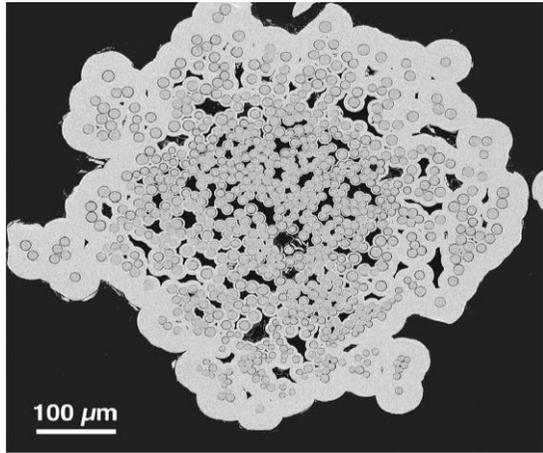


# Yarn Model Verification – Off-axis Tension

- Yarn model with matrix damage verified through off-axis tension simulations
- Tangent modulus of yarns during ‘constant matrix stress’ regime in good agreement with experimental observations



# Yarn Model Validation – Mini-composite Tests



## Yarn volume fractions

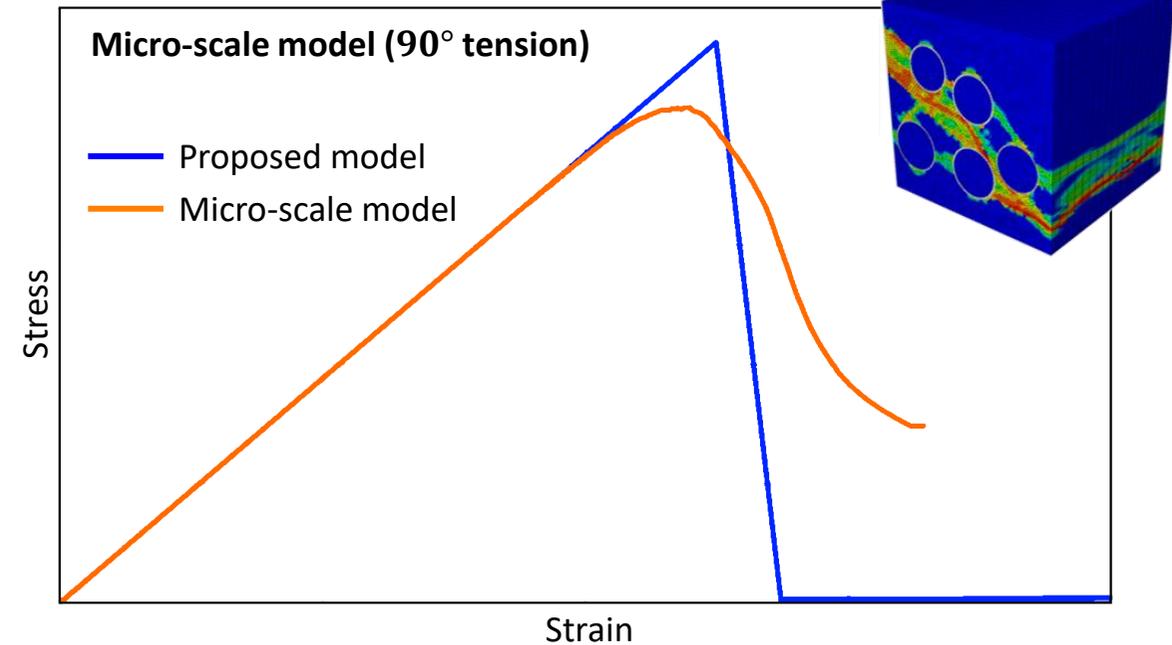
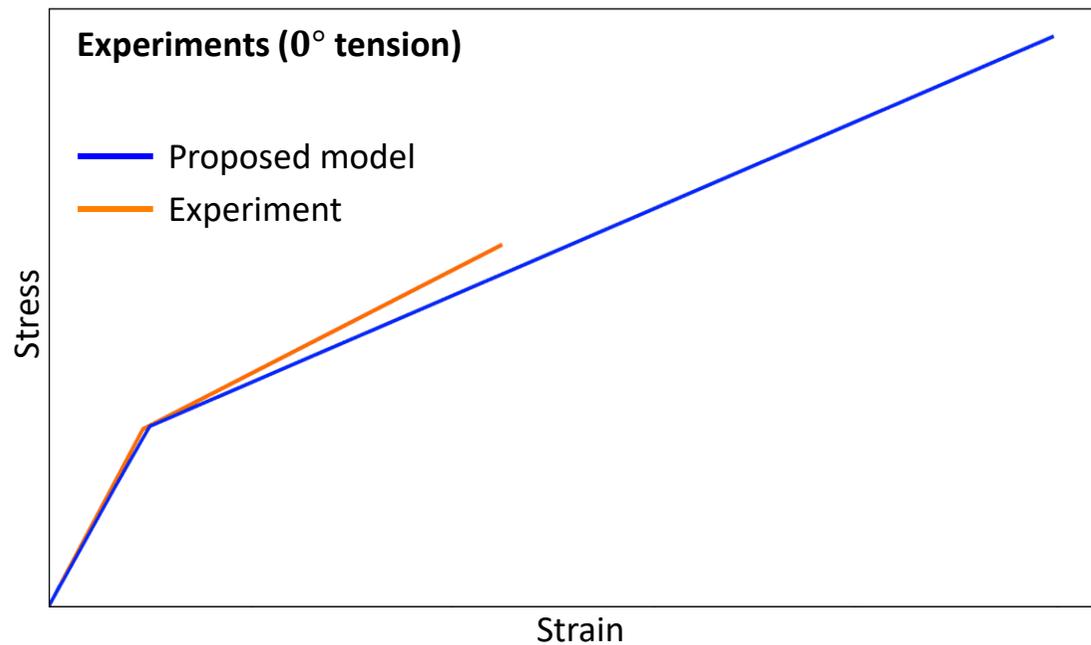
- Fibre - 25.8%
- BN - *Export Control*
- Void - *Export Control*
- CVI - *Export Control*

## Rule of Mixtures

- $E_{[\text{yarn matrix}]}$  - *Export Control*
- $\sigma_{[\text{yarn matrix}]}^{\max}$  - *Export Control*

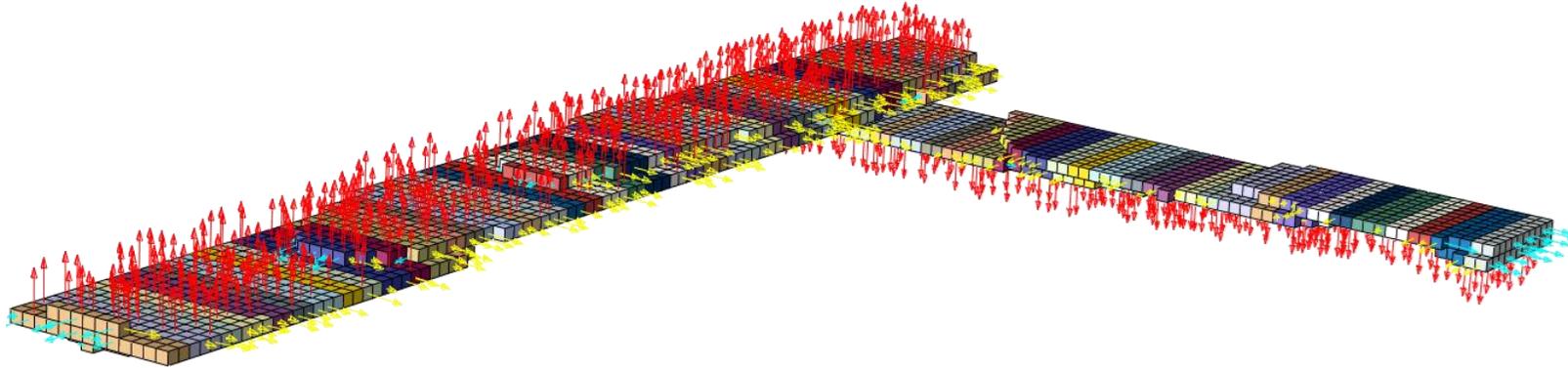
## Constituent properties (from literature):

| Constituent    | $E$<br>[GPa] | $\nu$ | $\sigma_{\max}$<br>[MPa] |
|----------------|--------------|-------|--------------------------|
| Fibres         | 300          | 0.17  | 3200                     |
| CVI SiC Matrix | 460          | 0.17  | 500                      |
| Boron Nitride  | 20           | 0.22  | -                        |
| SMI SiC Matrix | 299          | 0.17  | 170                      |

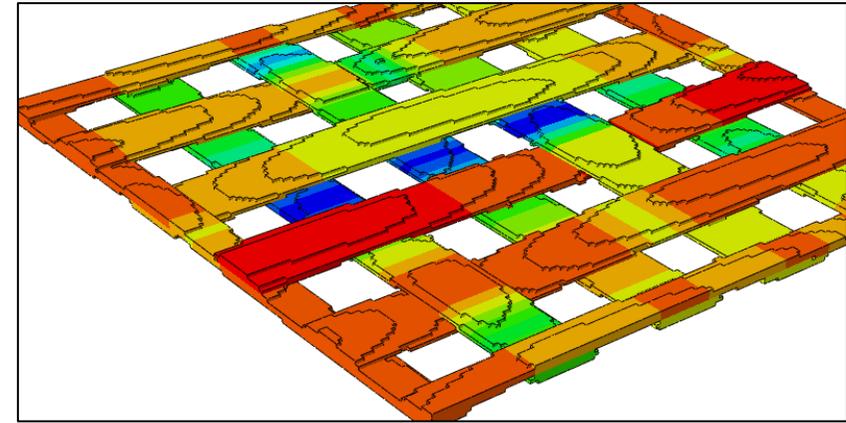


# Local Fibre Volume Fractions and Fibre Orientations

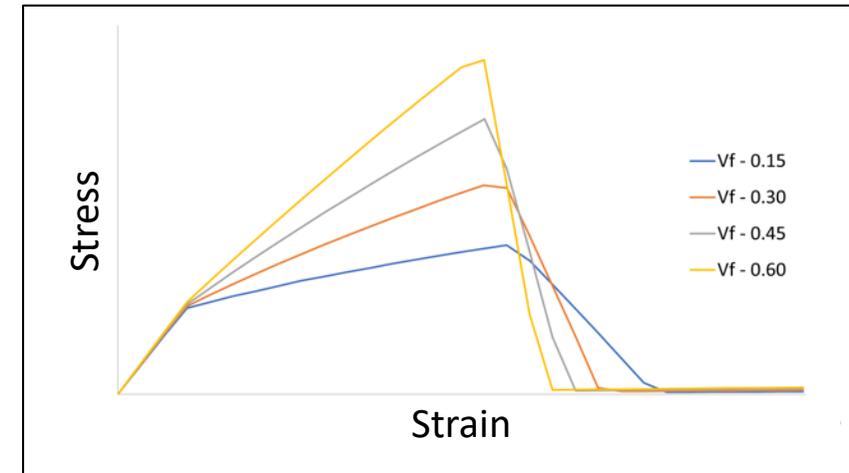
The Mori-Tanaka + CDM model is applied to individual yarns in the weave taking into account local material axes and fibre volume fractions obtained from weaving and compaction simulations (SimTex)



Individual yarns and their material orientations



Fibre volume fraction in yarn elements of meso-scale models



Comparison of different fibre volume fractions

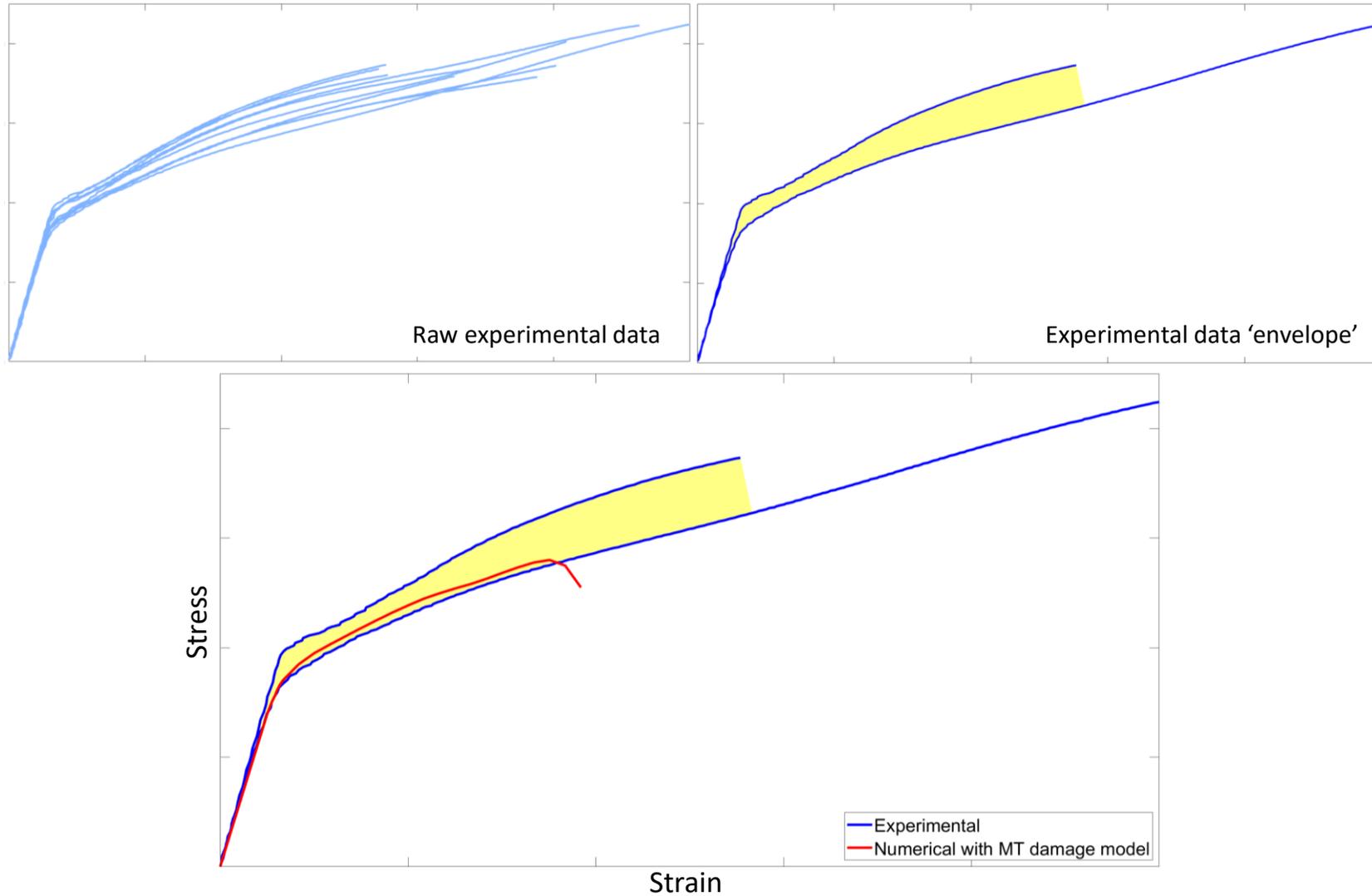


# Verification and Validation Test Cases

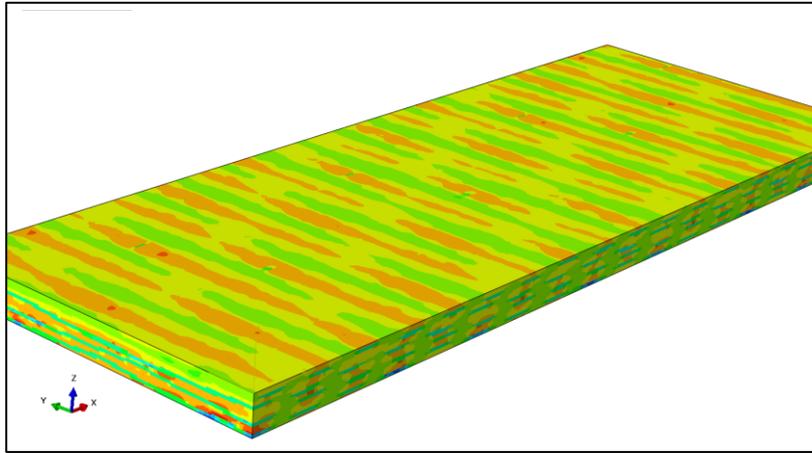
- Unnotched tension (R-R)
  - Gauge section dimensions: 27.94 mm × 10.5 mm
  - Data recorded: Force-displacement curves
- Unnotched tension (Swansea UTC)
  - Gauge section dimensions: 7.6 mm × 3.8 mm
  - Data recorded: Force-displacement curves, SEM images (centre of gauge length)
- Open-hole tension (R-R)
  - Proprietary dimensions
  - Data recorded: Force-displacement curves, DIC maps



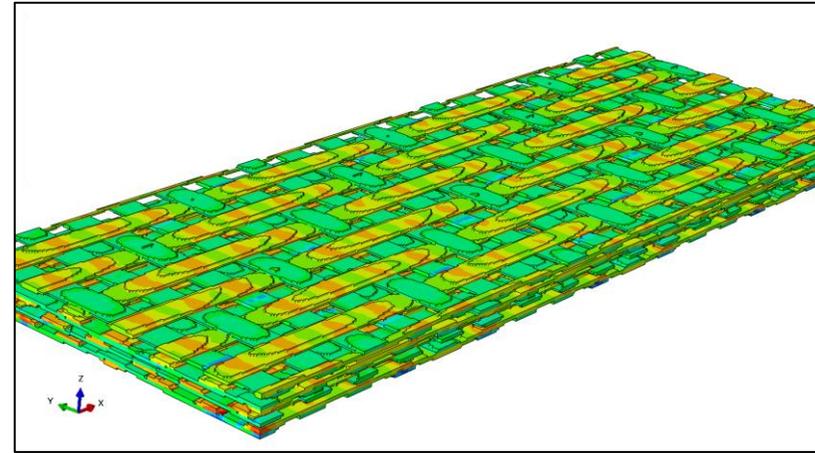
# Unnotched Tension Tests – Global Stress-Strain Response



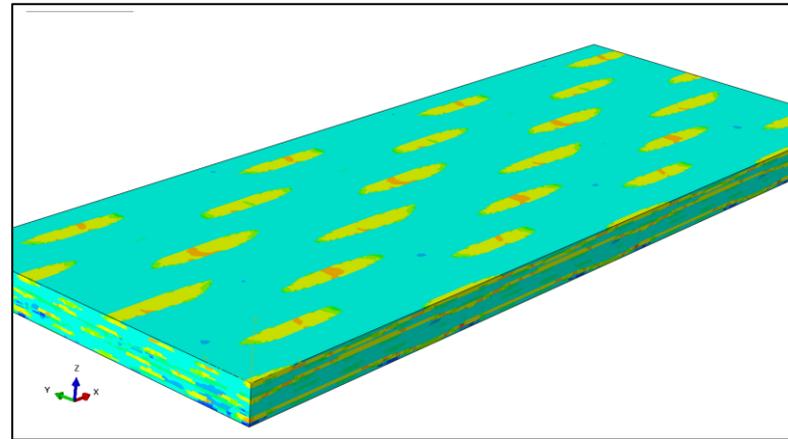
# Unnotched Tension Tests – Direct Stresses $\sigma_{11}$



Composite stresses, pristine



Yarn stresses, pristine



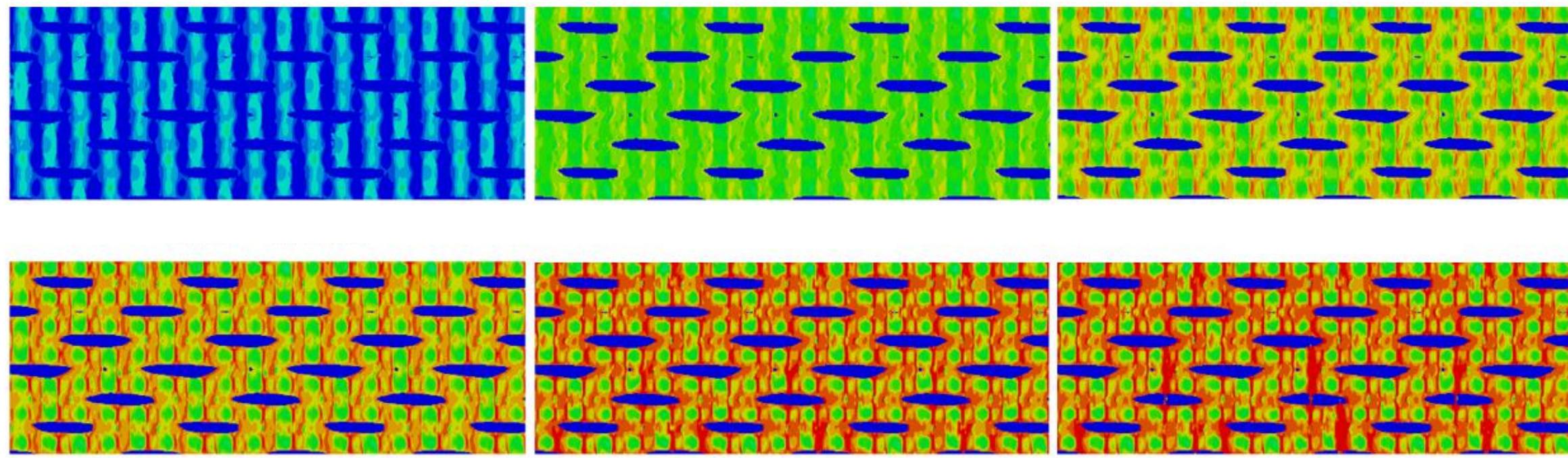
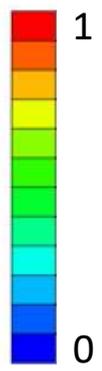
CMC textile stresses,  
after bulk matrix damage



# Unnotched Tension Tests – Damage Contours

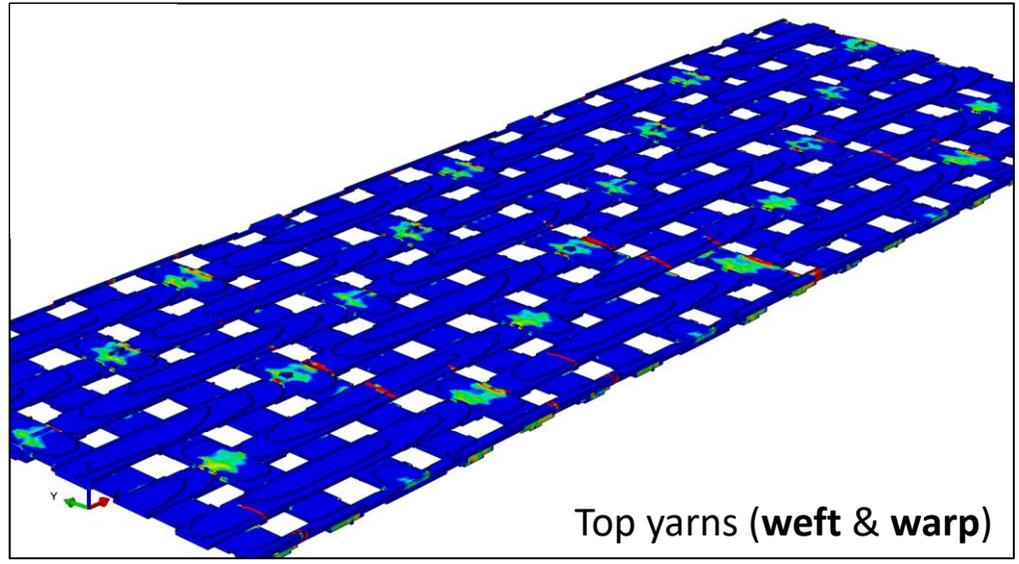
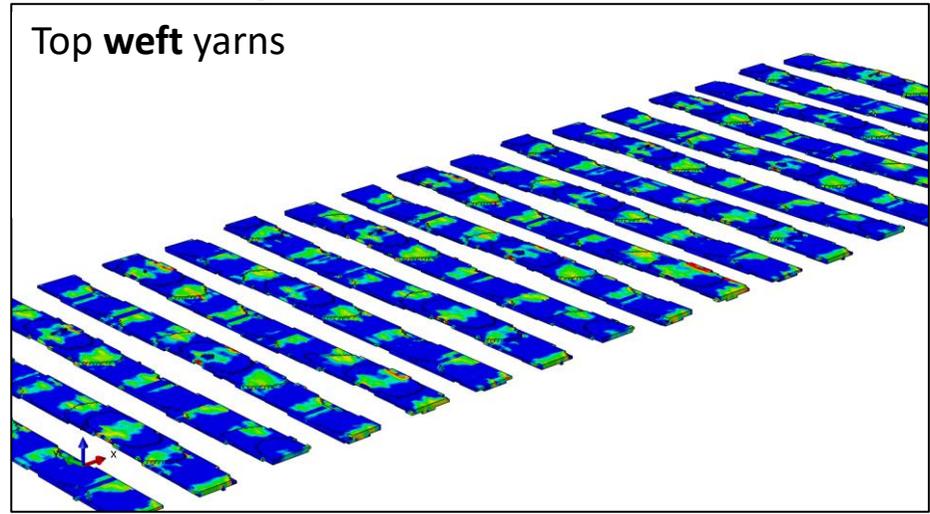
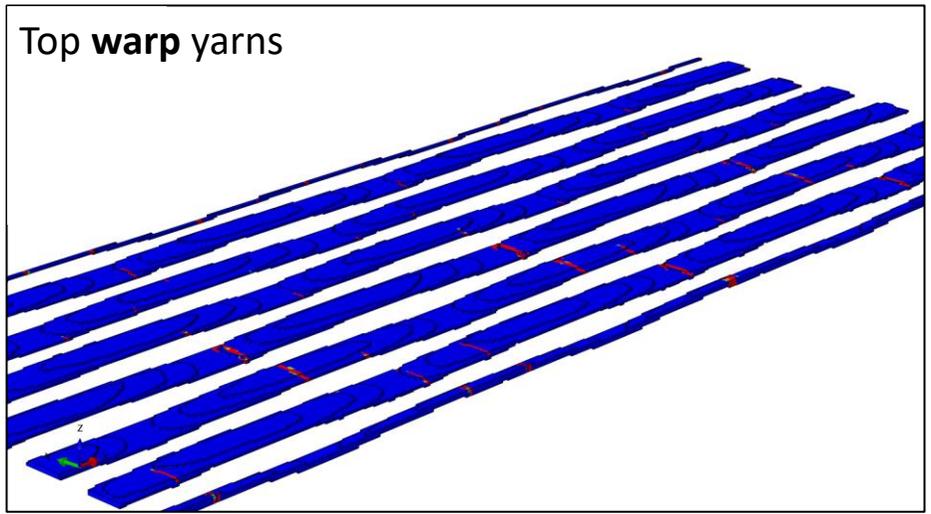
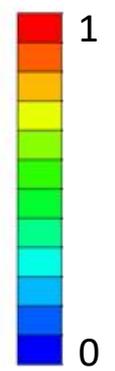
Increasing applied strain  
→

Damage variable



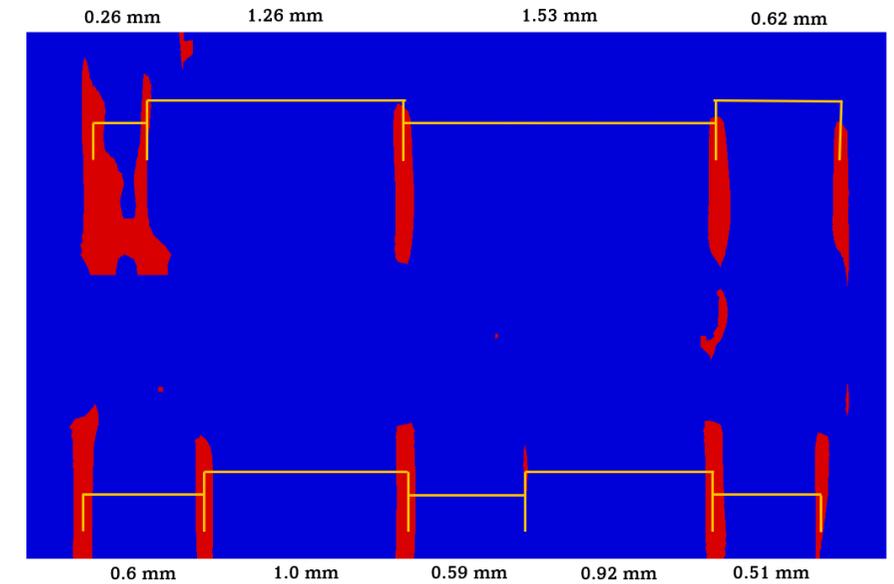
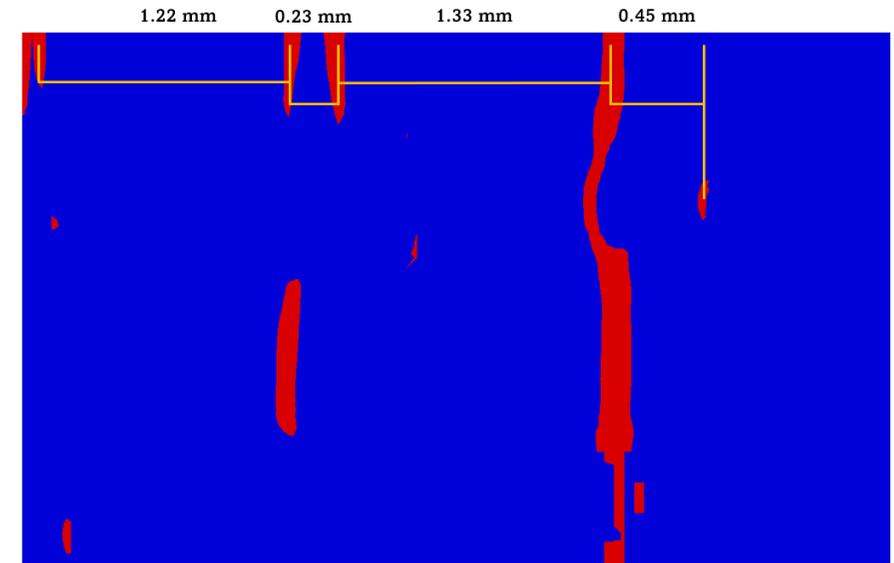
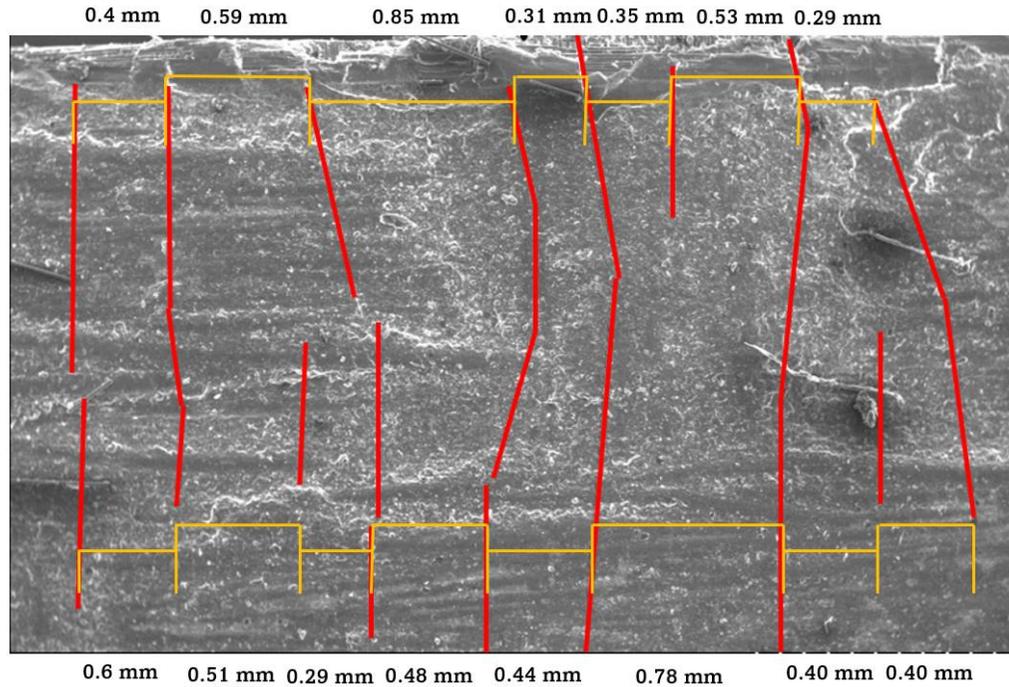
# Unnotched Tension Tests – Yarn Damage at Final Failure

Damage variable



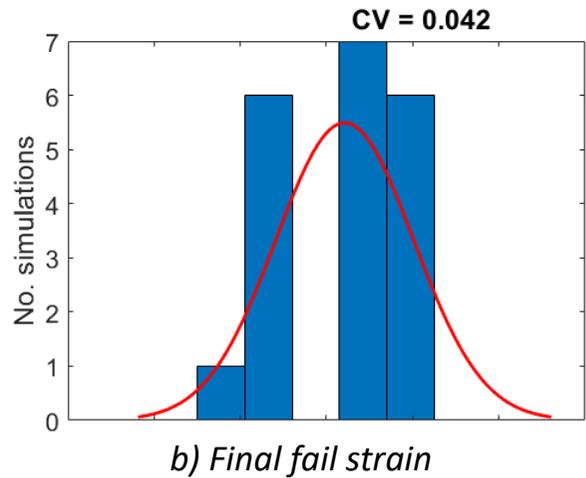
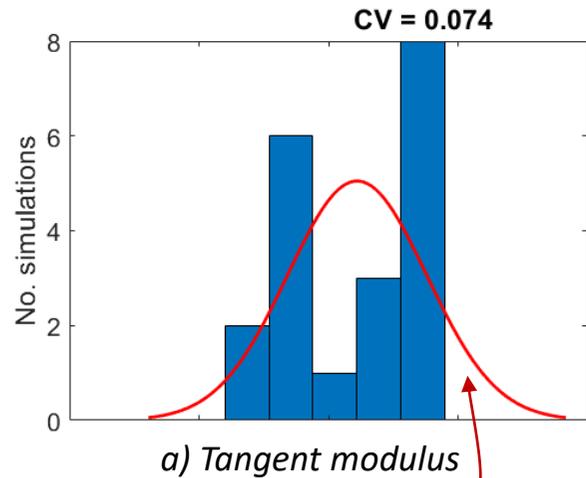
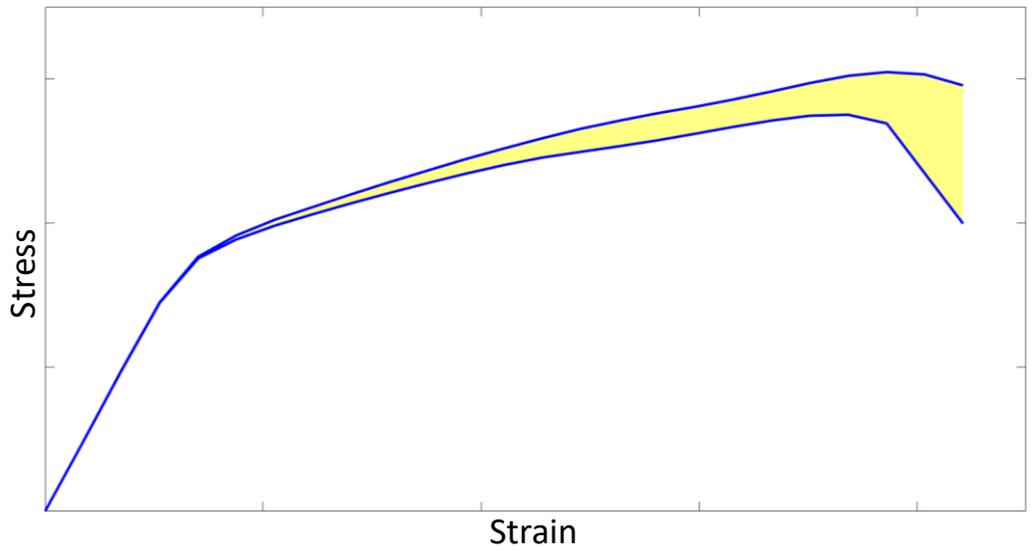
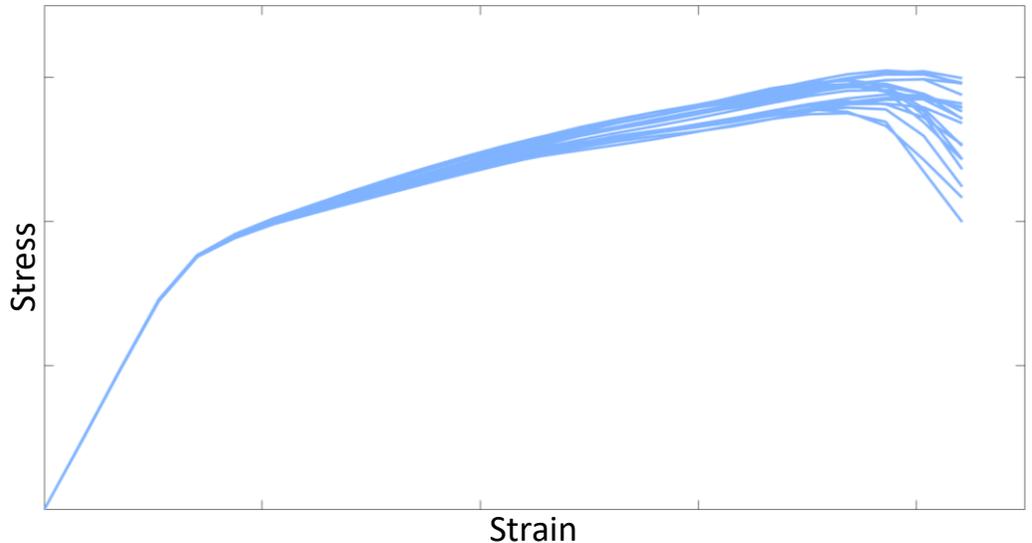
# SEM Evidence (Swansea UTC)

- Crack orientations and spacing measured by SEM at increasing loads and at final failure (shown here)
- Similar 'tram-track' shaped cracks in both cases
- Experiments show more closely-spaced cracks → models assume a homogeneous bulk matrix (without particles or pure Si)

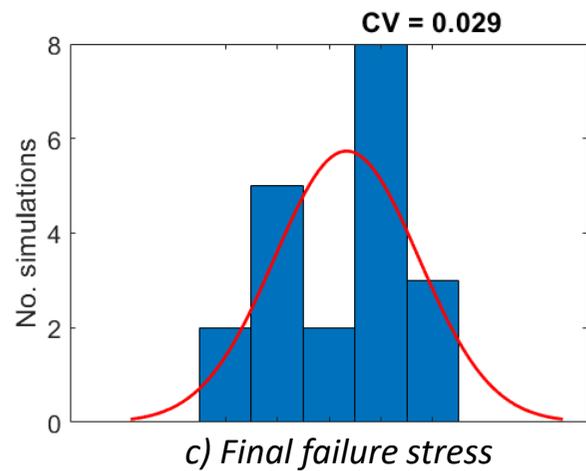


# Sensitivity Study – Random Yarn Shifting

Proposed modelling framework enables the study of **manufacturing variabilities** such as random ‘shifting’ between adjacent yarns

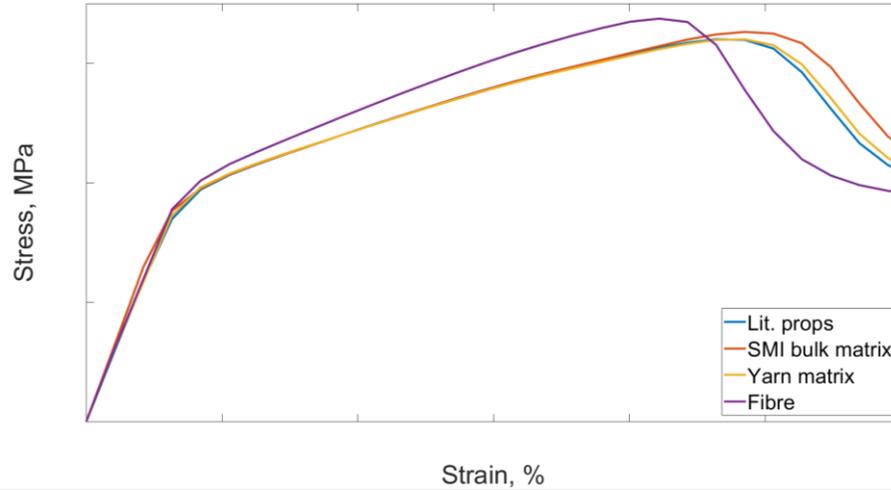


Assuming a normal distribution, for 99.8% of yarn shifting cases the tangent modulus varies within  $\pm 44.5\%$  of the mean value



# Sensitivity Study – Constituent Properties

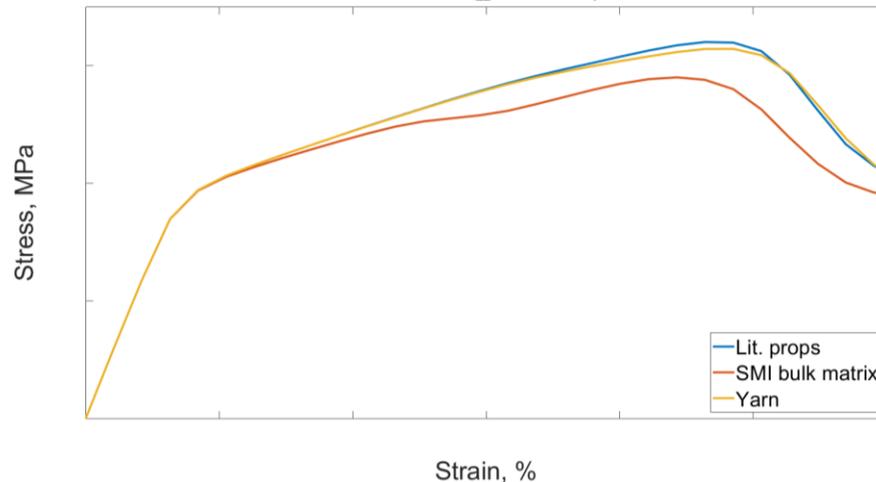
Modulus, +20%



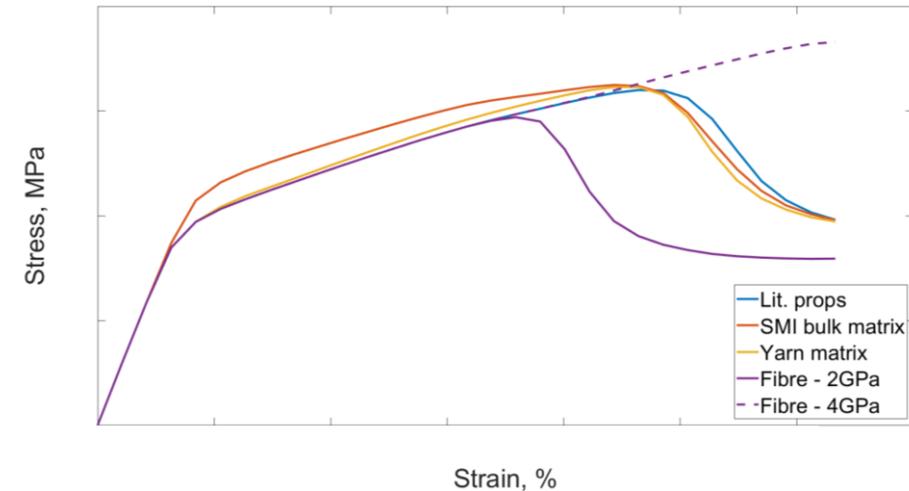
Proposed modelling framework enables the study of **material variabilities**

- SMI bulk matrix properties have a significant influence on the CMC properties
- Fibre modulus has a significant influence on CMC tangent modulus
- Variation in fibre strength properties reported in literature can change CMC failure strain by up to 100%

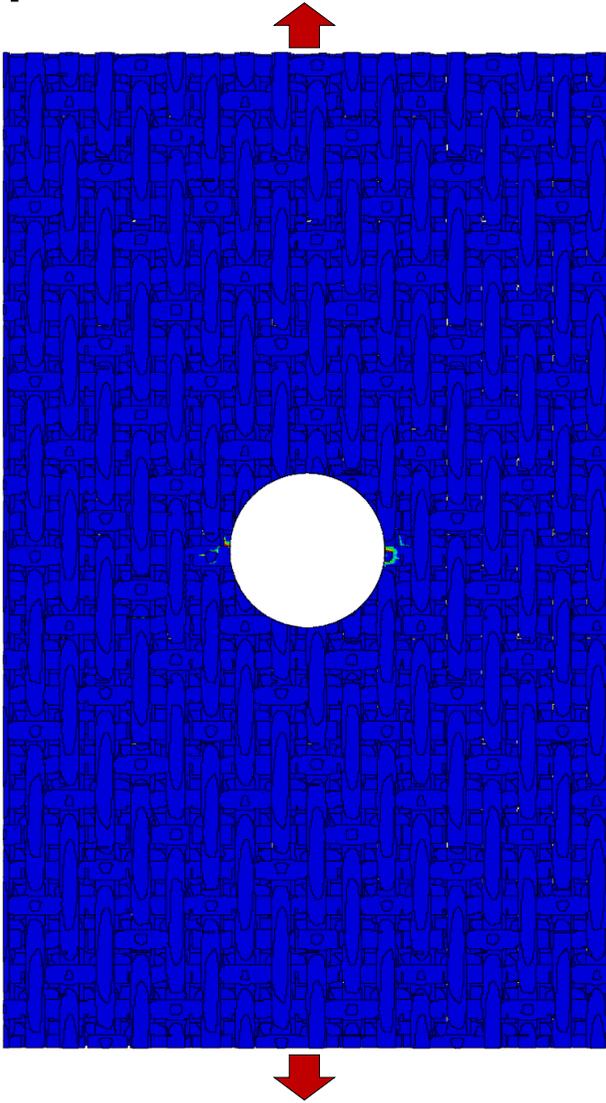
Fracture toughness, -50%



Strength, +20%

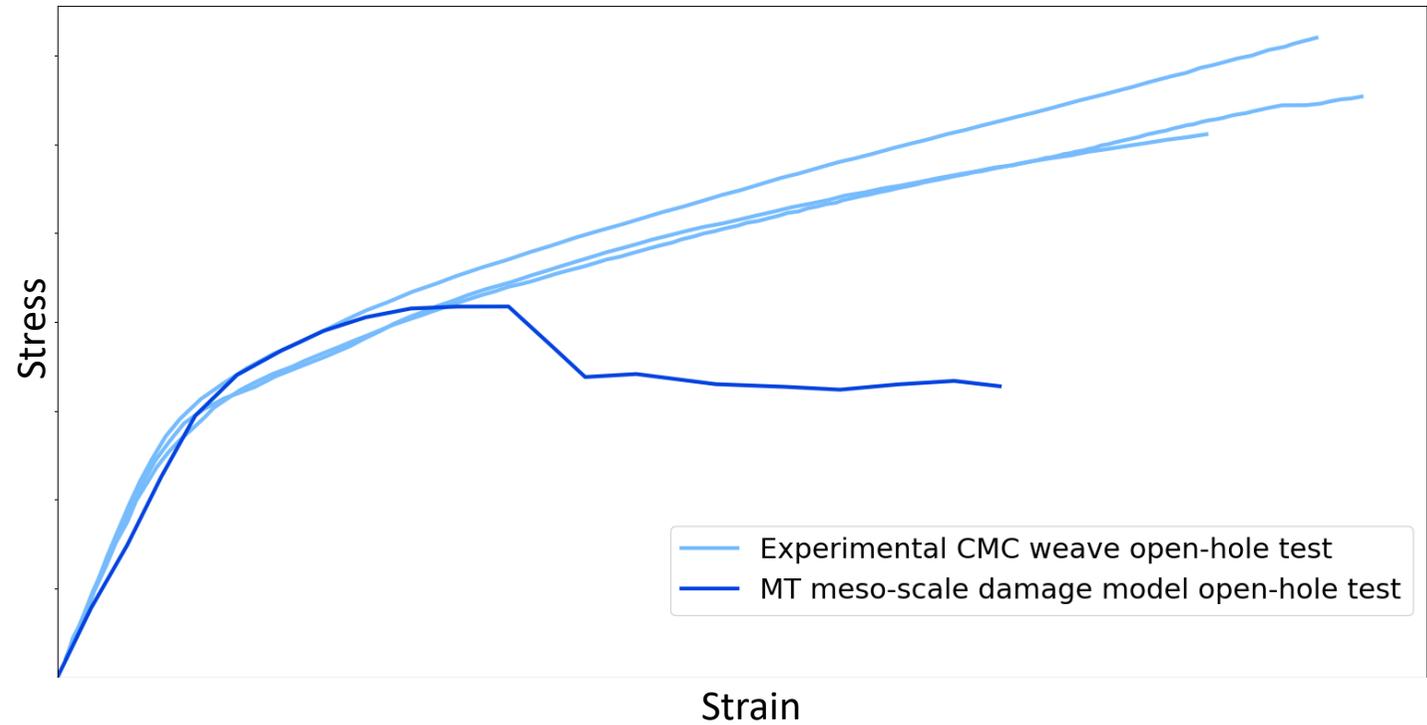


# Open-Hole Tension Tests



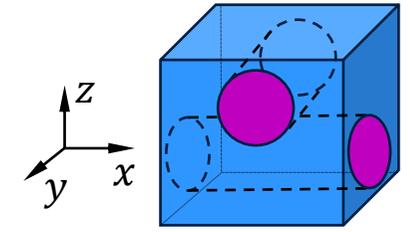
The modelling framework was also applied to feature-level open-hole tension tests

- Very large models ran on Bristol's HPC facilities
- For large applied strains the model overestimates the extent of damage, consistently with uniaxial tension results
- Open-hole tests believed to be highly sensitive to manufacturing variabilities

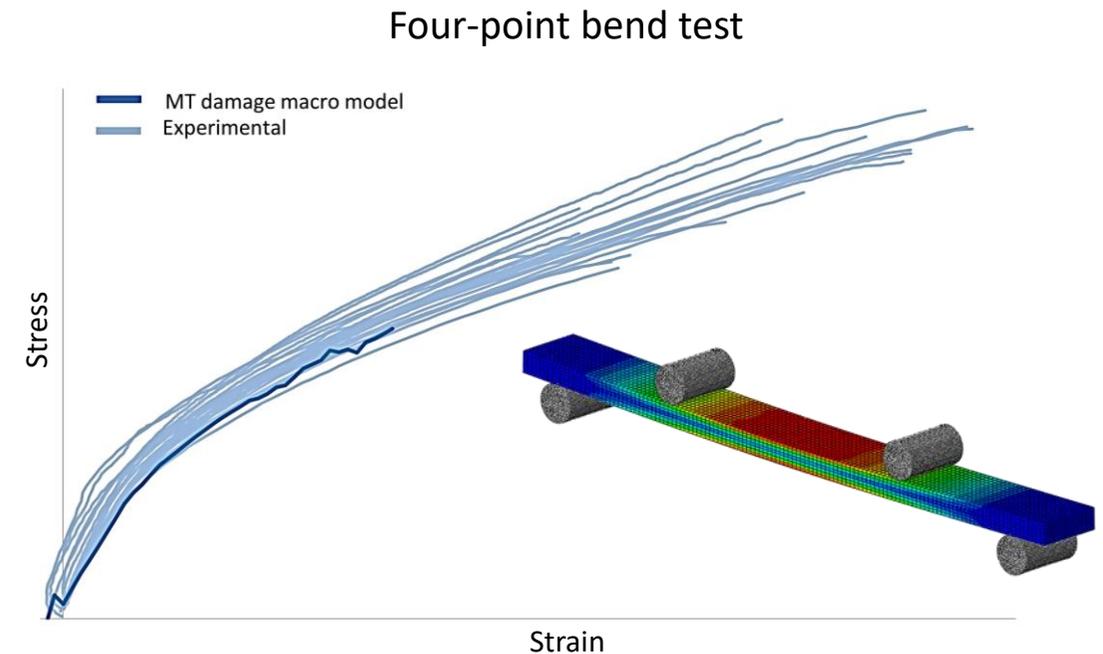
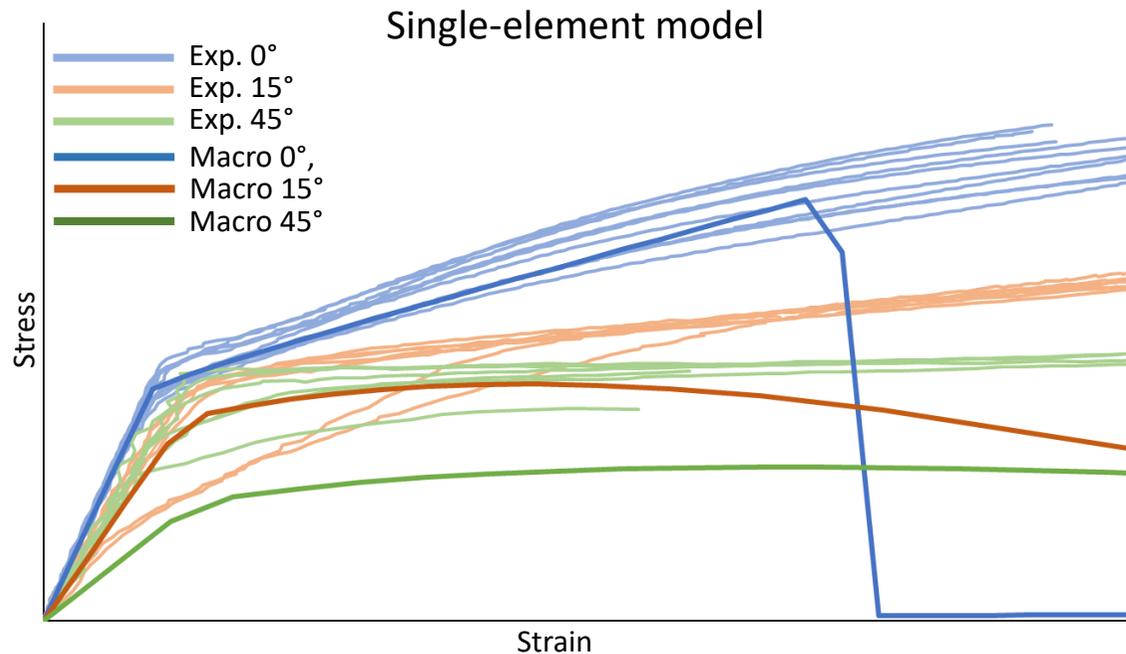


# Homogenised Macro-scale Material Model

- Homogenised macro-scale model suitable for component-scale analyses
- Fibre inclusions at  $0^\circ$  and  $90^\circ$  to simulate weave
- Underlying weave structure may be considered using fibre and interphase ( $0^\circ$  and  $90^\circ$ ) plus CVI and SMI volume fractions



Model extended to multiple fibres



# Summary

- A 3D meso-scale material damage model for woven CMCs was developed which takes constituent properties, volume fractions and fibre architecture as input
- Assumptions made to account for complex damage mechanisms and limited material data are suitable at this stage
- Combined with weaving & compaction models give good agreement with experiments

## Future work

- Further experiments required to validate meso-scale model
- Stochastic properties for bulk matrix to better match the real micro-structure
- Material model currently being extended to include thermal, environmental, creep and fatigue effects



# Acknowledgements

- The authors wish to acknowledge the support of Rolls-Royce plc through the Composites University Technology Centre (UTC) at the University of Bristol, UK.

